

# Surface Characterization for SRF

**Charles Reece** 

USPAS Course: SRF Technology: Practices and Hands-On Measurements

January 2015



C. Reece

# **SRF** Surfaces

### SRF cavity surfaces must be "pure", "clean", and "smooth"

- "Ideal" surface is defect-free Nb crystals with only Nb<sub>2</sub>O<sub>5</sub> ~4 nm capping layer and planar surface topography.
- After practical cavity fabrication, the real surface is "disturbed" and "polluted."
- Empirically found that >100 µm removal is typically required to reliably expose "good" bulk Nb material, i.e. predictable SRF performance.
- SRF cavity performance limitations always result from particular details of the surface.
  - Multipacting, field emission, quench, "Q-slope", "Q-drop", "Q-rise", Q<sub>0</sub>
- A lot of detailed attention is required to understand how to "grow" excellent SRF thin film surfaces.





### SRF is inherently a surface phenomenon

• RF Supercurrents flow only within a <u>very</u> shallow depth: penetration depth -  $\lambda$  ~40 nm for Nb

e.g., for a 9-cell Tesla-style cavity < 0.1 cm<sup>3</sup> of Nb actually matters for SRF

- To understand and control the desirable properties of SRF cavities requires knowledge and control of this thin layer of material over large surfaces
  - **Composition** elemental, structure, interstitials
  - Morphology pits, scratches, edges, topography
  - External contamination particulates, condensed gases





# **Structure Determines Properties**



### Microscopy & Microanalysis

Experimental methodologies which employs (electron-optical) instrumentation to spatially characterize matter on scales which range from tenths of a millimeter to tenths of a nanometer. The principle modalities employed are:

#### Imaging

Scanning Electron Microscopy Transmission Electron Microscopy Scanning Transmission Electron Microscopy Focussed Ion Beam

#### Diffraction

Electron Backscattered Difrraction Selected Area Electron Diffraction Convergent Beam Electron Diffraction Reflection High Energy Electron Diffraction

#### Spectroscopy

X-ray Energy Dispersive Electron Energy Loss Auger Electron



Jefferson Lab



# Elemental Analysis

- X-ray Energy Dispersive Spectroscopy (EDX)
- <u>Auger</u> <u>Electron</u> <u>Spectroscopy</u> (AES)
- X-ray Photoemission Spectroscopy (XPS)
- <u>Secondary</u> <u>Ions</u> <u>Mass</u> <u>Spectroscopy</u> (SIMS)

Material Science of Thin Films, Tutorials at JLab, Xin Zhao

6

Jefferson Lab

USPAS SRF Course Jan. 2015



#### <u>Scanning Electron Microscope (w/EDX) vs.</u> <u>Scanning Auger Microscope/Spectroscope (SAMs)</u>



An emitted Auger electron will have a precise kinetic energy  $E_k$  $E_k = E_{\text{Core State}} - E_B - E_C'$ Electron Auger electron, EKLL Hole 0 Vacuum Work function,  $\phi$ Fermi level Primary electron E beam, E<sub>KE</sub> X-ray Eκ Schematic of Auger Effect

7

Jefferson Lab



USPAS SRF Course Jan. 2015

# Auger Electron Spectroscopy



Because the Auger peaks are superimposed on a large continuously incremental background in direct Auger spectrum, the peak feature is not distinguished in the direct representation. The energy distribution spectrum N(E), is differentiated to enhance the peak features. Thus, the conventional Auger spectrum's representation is the function, dN(E)/dE. Material Science of Thin Films, Tutorials at JLab, Xin Zhao



Jefferson Lab

# AES and EDX Spectrum





Jefferson Lab

# **SRF** Surfaces

### Auger Emission Spectroscopy – AES

- <u>Very</u> surface sensitive: few monolayers
- Not commonly used
- Adsorbed gases are usually irrelevant to SRF performance

# Trouble-shooting sulfur surface contamination following EP

X. Zhao, et al. PRST-AB **13**, 124702 (2010), <u>http://link.aps.org/doi/10.1103/Phys</u> <u>RevSTAB.13.124702</u>



AES spectra of a spot before and after a slight sputtering. **Fewer than 10 atomic layers** were removed by the Ar<sup>+</sup> beam. The sulfur peak was greatly reduced after sputtering, which indicated sulfur coverage is ultra-thin.





# **SRF** Surfaces

### SEM/EDX – Scanning Electron Microscope with Energy

#### **Dispersive X-ray spectroscopy**

Probing depth: ~1 μm

SEM micrograph of one crystallite-like particle on Nb surface after EP.

The yellow box labels an area surveyed by EDX.



Photo Energy (KeV)

11

Jefferson Lab

X. Zhao, et al. PRST-AB **13**, 124702(2010) <u>http://link.aps.org/doi/10.1103/PhysRevSTAB.13</u> .124702 EDX spectrum surveyed from the yellow box area. It contains S, Fe, N element besides Nb and O. The origin of Fe and N are unknown. The elemental ratio of N, O, S, Fe and Nb is 19:72:5:3:1 after commercial EDAX software evaluation.



# XPS

- X-ray photoemission spectroscopy
  - Measure the binding energy of electrons reveals their atomic and chemical origin
  - Surface sensitive probe of chemical composition

- For fixed photon energy hv, measuring the kinetic energy (KE) of the ejected electrons leads to an energy balance: hv = KE + BE, where BE is the binding energy.
- Measuring the KE of ejected electrons gives their BE, a description of the electron structure.



#### Example:

#### Near-Surface Composition of Electropolished Niobium by Variable Photon Energy XPS

H. Tian et al. SRF2003





### Secondary Ion Mass Spectrometry (SIMS)



• Escape depth of sputtered species only few Angstroms

• All elements and isotopes measurable (including H)

• ppm to ppb detection limit

• **10-20 n**m depth resolution typical, **1-2nm** at low energies

13

Jefferson Lab

R. G. Wilson, F. A. Stevie, C. W. Magee: Secondary Ion Mass Spectrometry, Wiley-Interscience (1989)



jlab.org

USPAS SRF Course Jan. 2015

# Improving SRF with "Pollution"

- "Pollution" of near surface discovered to have beneficial effect
- "High-Q<sub>0</sub>" by N-doping or Ti-doping
- What is going on here?





# XFEL/ILC recipe vs. N doping





USPAS SRF Course Jan. 2015



# Nitrogen Doping Process



# JLab HT-N treatment and triple single cell HEP configuration





### What does N treatment do? N depth profiles by SIMS



# **Crystallography Measurements**

- X-ray Diffraction (XRD)
- Electron Backscattering Diffraction (EBSD)
- Transmission Electron Microscopy (TEM)

Diffraction measurements probe the atomic structural patterns in the surface material

SRF requires high-quality lattice structure

Material Science of Thin Films, Tutorials at JLab, Xin Zhao





# **Crystal Quality**

# a definition based on crystallography

**CRYSTAL QUALITY** 

The quality of what is nominally a "single" crystal can vary over an enormous range. At one extreme, the crystal may have undergone gross plastic deformation by bending and/or twisting, such that some portions of it are disoriented from other portions by angles as large as tens of degrees, and the dislocation density is high. At the other extreme, some carefully grown crystals are almost free of dislocations and other line or planar imperfections, and their crystal planes are flat to less than  $10^{-4}$  degrees over distances of the order of a centimeter. In general, metal crystals tend to be more imperfect than crystals of covalent or ionic substances.

Various x-ray methods of assessing crystal quality are described below. These methods differ in sensitivity, and we will deal with the least sensitive first.

• Ref. "Elements of X-ray Diffraction", B.D. Cullity. 2<sup>nd</sup> edition, page 260.





# **Misorientation Angles**

### - as one caliber of crystal quality



Fig. 8-27 Reflection of white radiation by bent and polygonized lattices (schematic).

 Misorientation Angles of a survey area could be measured by <u>XRD</u> (Rocking Curve, RSM), or by <u>EBSD</u>





# XRD vs. EBSD

|   | XRD   | EBSD  |
|---|---|---|
| Probing Area<br>(Diffraction<br>Area)   | 10*17 mm<br>(selectable by X-ray<br>aperture) | <ul> <li>30*30 nm</li> <li>By rastering e-beam, can scan a large area</li> <li>Scanning area is limited by SEM magnification</li> </ul> |
| Probing Depth<br>(Diffraction<br>depth) | 1 - 2 μm                                      | < 50 nm   |
| Pole Figures                            | Yes   | Yes   |
| Grain size<br>sensitivity               | any   | Must > 50 nm  |





#### XRD Pole Figure Experimental Setup and Standard Nb (110) Pole Figure

Nb (110) Pole Figure





**Experimental Steps:** 

•Fixed 2 $\vartheta$  of a {hkl} crystal plane. (Bragg Law  $2d_{\{hkl\}}*sin(\vartheta)=\lambda$ ) •Rotated around Normal Direction (Azimuthal  $\varphi$ , from 0-360°) •Titled off-angle from Normal Direction ( $\psi$ , 0-90<sup>0</sup>)

P.F. is to visualize **Reciprocal Lattice Space** One **Crystal Plane stacks** in real lattice space is a **Pole** in reciprocal space X.Zhao et al, Talk on 5th SRF Thin Film Workshop, JLab. 2012

| Crystal Plane | $\psi (^{0})$ | $\varphi$ ( <sup>0</sup> ) |
|---------------|---------------|----------------------------|
| (110)         | 0             | 0                          |
| (011)         | 60            | 54.74                      |
| (101)         | 60            | 125.26                     |
| (1,0,-1)      | 60            | 234.74                     |
| (0,1,-1)      | 60            | 305.26                     |
| (1,-1,0)      | 90            | 180                        |
| (-1,1,0)      | 90            | 0                          |

Jefferson Lab





# Grains Orientation Mapping by EBSD

- Electron Beam Spot is small (<u>few nanometers</u>)
- By rastering electron beam on a sample to map grain orientations





- Pattern matching and decomposition
- Computationally intensive
- But implementations are now quite fast

24

USPAS SRF Course Jan. 2015



### EBSD – BCP on fine grain, nano-polished niobium

# Technique used to examine surface evolution during processing

6 minutes BCP at room temperature











### EBSD – BCP on fine grain, nano-polished niobium

### 6 minutes BCP at room temperature



C. Reece

#### A Historical Time Line in Electron Optical Instrumentation

- 1897 JJ Thompson Discovery of the Electron
- 1926 H. Bush Magnetic/Electric Fields as Lenses
- 1929 E. Ruska PhD Thesis Magnetic lenses
- 1931 Knoll and Ruska 1st EM built
- 1932 Davisson and Calbrick Electrostatic Lenses
- 1934 Driest & Muller EM surpases LM
- 1939 von Borries & Ruska 1st Commerical EM~ 10 nm resolution



- 1965 ~ 0.2 nm resolution (Multiple Organizations)
- 1968 A. Crewe U.of Chicago Scanning Transmission Electron Microscope ~ 0.3 nm resolution probe - practical Field Emission Gun

Ruska etal - Nobel Prize

1999 < 0.1 nm resolution achieved (OÅM ) $2009 \quad 0.05 \text{ nm} \text{ (TEAM)}$ 







27

USPAS SRF Course Jan. 2015

Jefferson Lab



# Service of Traditional Electron Microscope



### Morphology, Crystallography, Elemental, Chemical, Electronic Structure

Material Science of Thin Films, Tutorials at JLab, Xin Zhao

28



USPAS SRF Course Jan. 2015



#### Transmission Electron Microscopy













29

C. Reece

1/25/2013

Jefferson Lab

# **Traditional TEM Specimen Preparation**

hole

#### Foils

3 mm diam. disk very thin (<0.1 - 1 micron - depends on material, voltage)

- mechanical thinning (grind)
- chemical thinning (etch)
- ion milling (sputter)

examine region around perforation





### Sample Prep for TEM/STEM studies

TEM: high spatial resolution but sample has to be thin for HRTEM thickness < 50nm

Focused Ion Beam: vertical SEM column + Ga ion column + micromanipulator + gas injection system + detectors



Result: cross-sectional cut from the bulk

# Nb-H Superlattice at 94 K



### HRTEM imaging of Cold Spot at Room T continue



#### Phase Contrast of the Grain Boundary



#### No significant oxidation along Grain Boundary in contradiction to J.Halbritter (2001)

Dualing in any Transmission Flasher Mission and in a sing regults and hat leads when the Vulie Transition 7th CDF

# **Topography characterization**



| erface                           | A BEAUTINE            | No. 8 Concession, No. 8   | THE OWNER WHEN THE OWNER |   |                                |
|----------------------------------|-----------------------|---|---|---|--------------------------------|
| led 1                            |                       |   |   |   |                                |
| Progress Bar                     |                       | C:UsersVizhao/DocumentsVi.Z/L<br>enerie 73 76 /Ba.de.A.RIV<br>C:UsersVizhao/DocumentsV.Z/L.a.<br>C:UsersVizhao/DocumentsV.Z/L.a.<br>C:UsersVizhao/DocumentsV.Z/L.a. | aser project/LZ data/AFM 201305<br>Ner annel 21/201 mm. 4 25cm<br>ser project/LZ data/AFM 201309  | Jefferson L   |                                |
|                                  | 5                     | - Rut   | ming  | Reset Save Rep                                      | ort If Yellow?                 |
|                                  |                       |   |   | edi   |                                |
| í                                |                       | 1   | 179.5773  | 5.2892  |                                |
|                                  | <u> </u>              | AFM 5   | 163.6924  | 5.2835  |                                |
|                                  |                       | AFM 10X<br>WLI 230<br>WILI 124<br>WILI 124  | 512   | 25000   | -                              |
| D <sup>-6</sup> 10 <sup>-4</sup> | 10 <sup>-3</sup> 1    | ν <sup>2</sup> 10 <sup>-1</sup>   | 0 order<br>1st order<br>Zind order<br>3rd order   | Non Windowed<br>Biackman Windowed<br>Tukey Windowed |                                |
| Fitting                          | Congur X (Succinit I) |   | RMS from 1D PSD   | 1000  |                                |
| 2e-05                            | 9e-05 0.0003          | FEE77   | 154.2839  | 0.0012543   |                                |
| 9e-05                            | 0.0002 0.002          | RMS from 2D PSD<br>RMS 1D RMS 2D  | Power Ratio   | Running   |                                |
| + +                              |                       | 1   |   | 500 0   | 0 500                          |
|                                  |                       | .8  | 500   | L   | _ untitled fit 1<br>z vs. x, y |
| i.                               | •                     | .7  |   |   | $\supset$                      |
|                                  | +                     | 1.6   | ê 0   |   | - · ·                          |
|                                  | *                     | 1.5   | 19 🔦  |   |                                |
|                                  | *                     | 9.4   | 튶 -500  |   |                                |
|                                  | <b>*</b>              | 1.3 -   | ž   |   |                                |
|                                  | A.                    | 12  | -1000   |   |                                |
|                                  | · ·                   |   | 600 AOD   | ~~~~~   | 14. C                          |
|                                  |                       | 1.1   | 400   | 300   | 300 400                        |
|                                  | ·····                 |   |   | 200 200   | 300                            |
| 10"                              | 10' 10'               | 10 0.0 0.4 0.6  | 0.8 1   | 100 100 200   |                                |

C. Reece

- Hirox optical microscope, Phenom SEM
- Atomic force microscopy (<u>AFM</u>)
  - Tapping mode
  - <u>RMS roughness</u> (*R<sub>q</sub>*), height variation of peaks/valleys
- Power spectral density (<u>PSD</u>) of surface height
  - Customized program
  - 2<sup>nd</sup> order detrending
  - Blackman window
  - Width variation of peaks/valleys
  - Quantitatively describe sharp features





# Case 3:BCP on Mechanical Polishing



Typical surface finishes AFM scan 100µmby 100µm **PSD Structure changes! Why?** 



#### Optical, SEM, AFM - Bubble prints, BCP on bi-crystal niobium





BCP 20°C, 12 minutes Print radius ~ 50 μm Print depth ~ 1 μm

C. Reece





USPAS SRF Course Jan. 2015



### Not all Nb "EPs" the same

With "standard"1:10 HF/H<sub>2</sub>SO<sub>4</sub> Electrolyte at 30°C Nb crystallography affects the polishing effectiveness.

With identical starting topography from CBP, given identical 100 min "EP" at 30°C, single-crystal material was significantly smoother.

Evidence for a significant etching activity at 30°C









### AFM, PSD - EP topography vs. surface flow rate

#### 14 V, 20-22 °C, 90 minutes, ~40 µm removed







# Importance of topography





C. Xu, C. E. Reece and M. J. Kelley, "Simulation of non-linear SRF losses derived from characteristic Nb topography: comparison of etched and electropolished surfaces," *http://arxiv.org/abs/1406.7276*, 2014.

EP cavities often have higher field gradients Difference between BCP and EP: topography

39



USPAS SRF Course Jan. 2015



# SRF is all about Surfaces

- Alphabet soup of analytical techniques available (short list)
  - Elemental analysis
    - EDX (bulk), AES, XPS, SIMS (surface sensitive)
  - Structural analysis
    - XRD (1-2 μm), EBSD (50 nm), TEM (~1 nm)
  - Topographical analysis
    - Profilometer, AFM
- RF measurements are always averages over large surface areas – often ambiguous interpretations



