

# DFT Analyses of Structures that Limit SRF Cavity Performance

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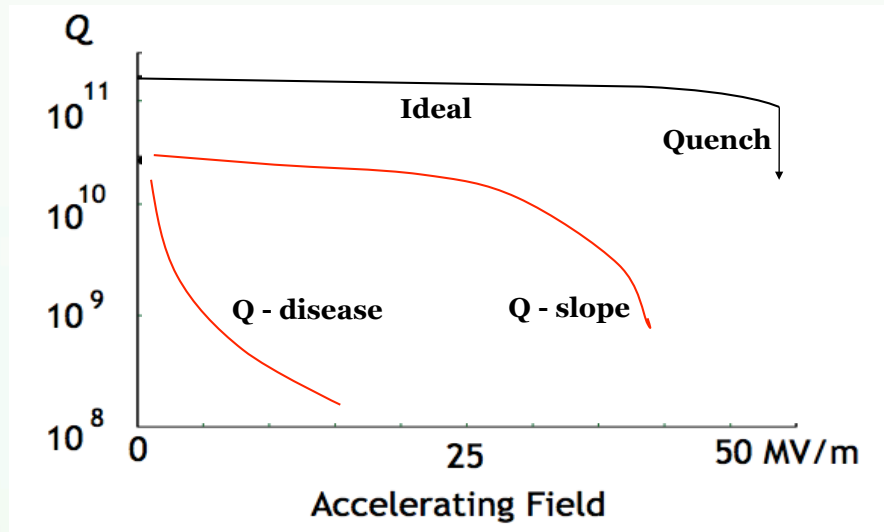
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# Overview

- Background - Impurities in niobium SRF cavities
- Methods - Modeling crystal structures with DFT
- Hydrogen in Niobium
  - Niobium hydride phases
  - Interactions with the ideal and defective lattice
  - Application to SRF cavity performance and processing
- Other impurities (O, N, C) in Niobium
  - Niobium Oxide phases
  - Effect on hydride phase formation
  - Application to SRF cavity performance and processing
  - An additional concern

# Superconducting Radio-Frequency (SRF) Cavities

## Limitations Related to Impurities in Niobium



- **Impurities** can
  - be dissolved in the metal and cause reduction of  $T_c$  and local heating
  - form precipitates with local magnetic moments or reduced  $T_c$

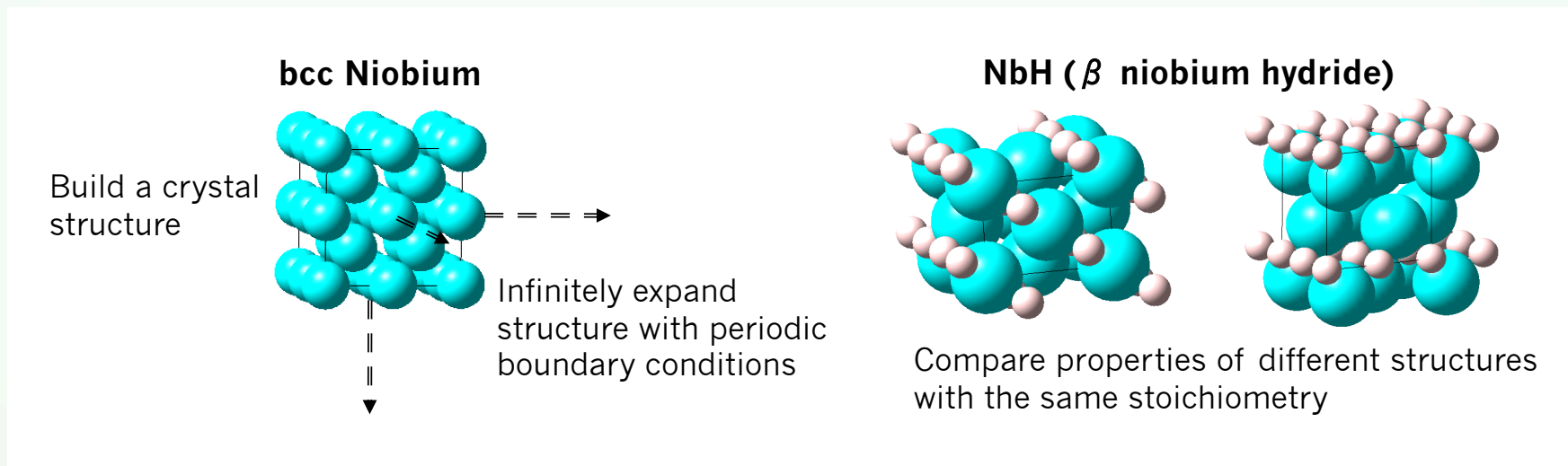
- Resulting cavity behaviors include
  - **Q - disease**, caused by large hydride precipitates?
  - **Q - slope**, caused by smaller precipitates; dissolved oxygen, hydrogen, nitrogen, etc.; magnetic structures?

# Impurity Absorption into Niobium Cavity Forming and Processing Procedure

- Forming
  - Complex – shape and ultra high purity – source of many lattice defects
- Processing
  - **Bulk electropolish** (~150  $\mu$  m) of inner surface - remove damage layer from forming
  - Wash - remove chemical residues from EP
  - **800 °C bake** - outgas hydrogen absorbed during chemical processing
  - Tune - shape changes during processing
  - **Fine electropolish** - smooth surface
  - Wash - remove chemical residues from EP
  - Rinse - remove dust (prevent field emission)
  - Assemble
  - **120 °C bake** - it works (why?)
  - Test performance

# Density Functional Theory Modeling

## Creating a Model



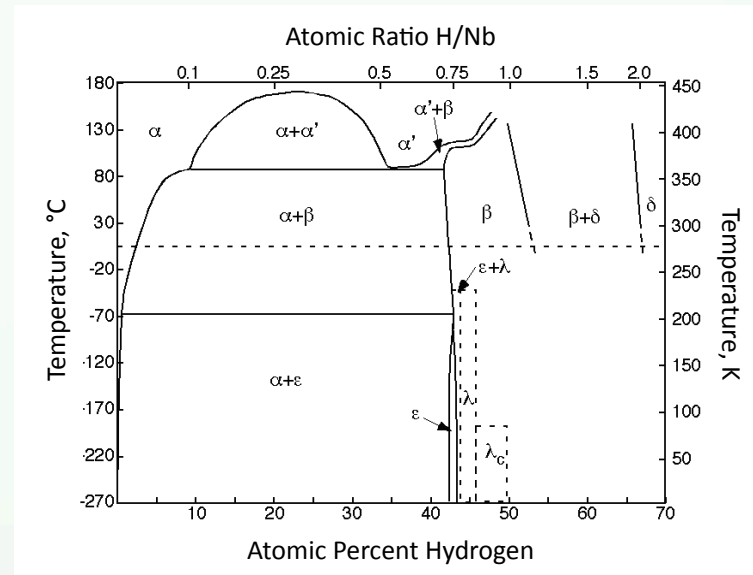
- Properties of interest include:
  - Energies – electronic, vibrational, strain
  - Geometry
  - Charge distribution
  - Magnetic moments
  - Phonons

# Density Functional Theory Modeling Calculations

- Solve the electronic structure problem for the model system
- Hohenberg, Kohn, Sham – obtain properties from electron density instead of many-electron wavefunction:  $E = T(\rho) + V(\rho) + \text{Exc}(\rho)$ 
  - $T(\rho)$  – kinetic energy
  - $V(\rho)$  – potential energy
  - $\text{Exc}(\rho)$  – exchange-correlation energy - unknown form  
-> describe with parameterized ‘functionals’
- Parameters
  - Vienna Ab Initio Simulation Package (VASP)
  - Plane wave basis set w/400 eV cutoff
  - PAW pseudopotentials to describe atomic cores
  - PBE-GGA exchange-correlation functional
  - 0.25/Å gamma-centered Monkhorst-Pack  $k$ -point mesh
- Bader Method to assign local properties

# Hydrogen in Niobium

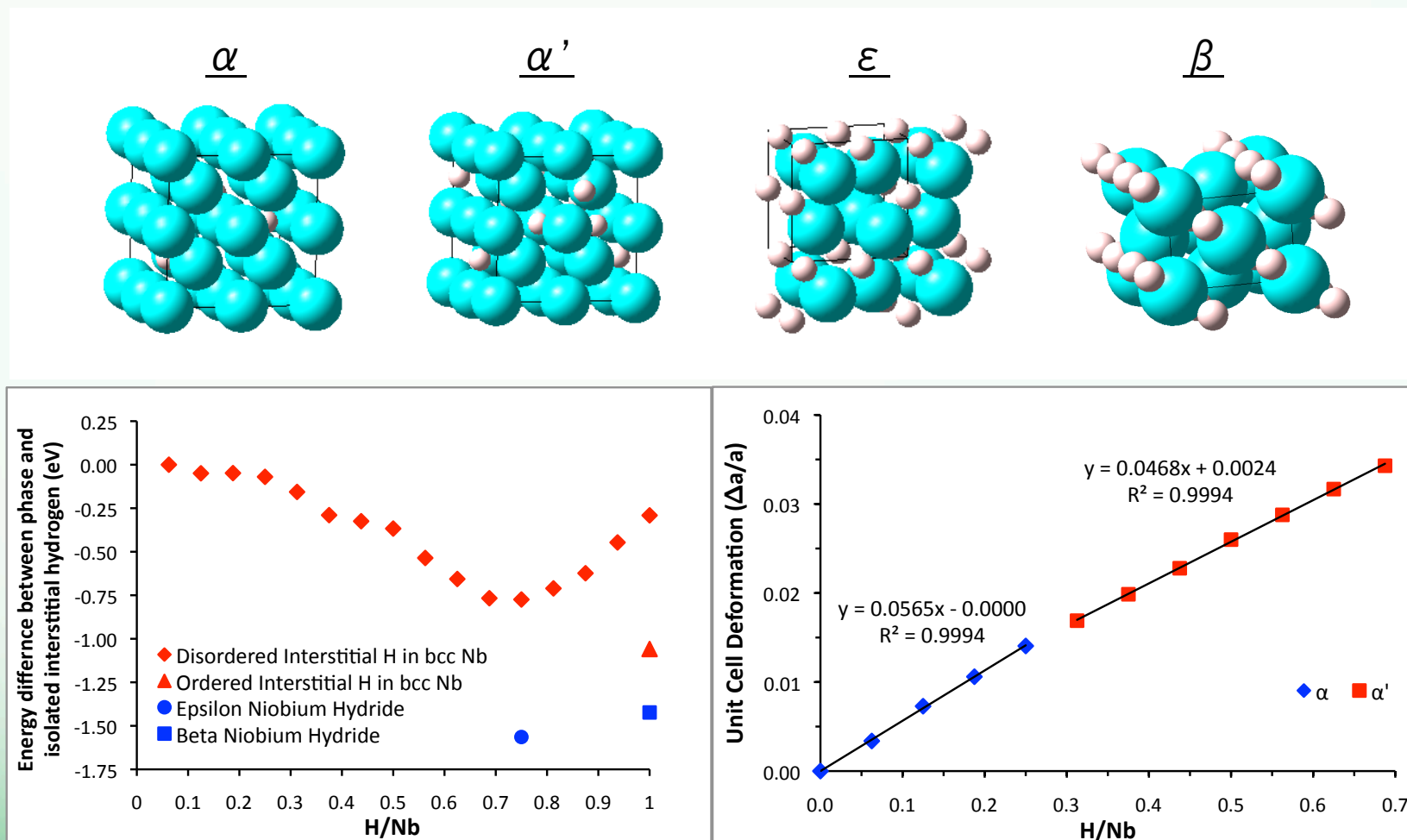
- The niobium – hydrogen phase diagram is very complex
  - $\alpha$ ,  $\alpha'$  – interstitial hydrogen dispersed in bcc niobium
  - $\beta$ ,  $\varepsilon$  – ordered hydrogen interstitials in fcc niobium
  - $\delta$  – ordered hydrogen interstitials in fcc niobium – fluorite structure
  - $\lambda$ ,  $\lambda_c$  – experimentally unconfirmed phases



R.E. Ricker, G.R. Myneni, J. Res. Natl. Inst. Stand. Technol., 115, 1 (2010).

- What do we have in SRF cavities and how is it affected by processing?

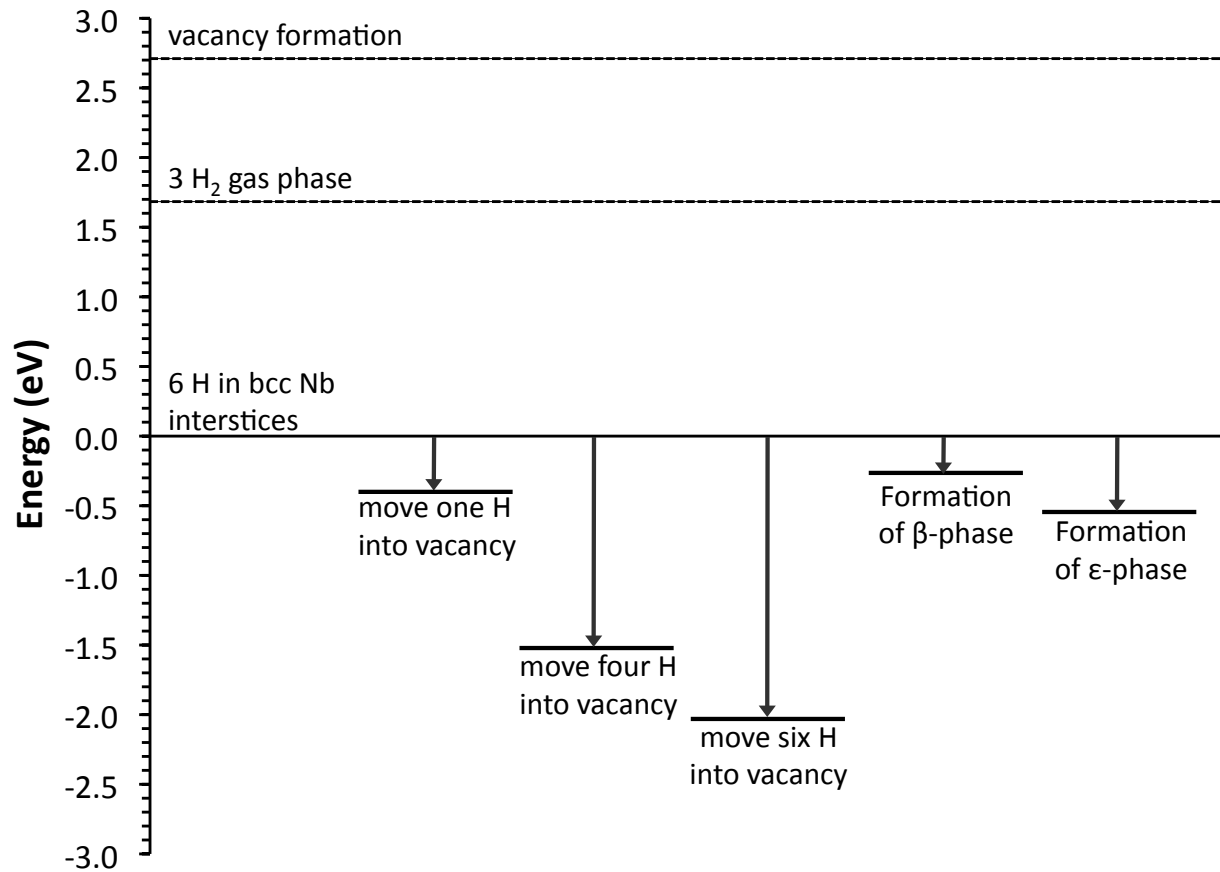
# Hydrogen in Niobium Phase Models





# Hydrogen in Niobium

## How do Phases Form?

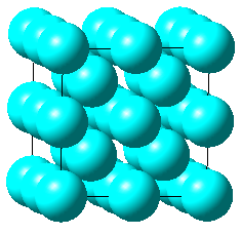


# Application to SRF Cavities

- Hydrogen in niobium could be a source of Q-slope and Q-disease.
  - Ordered hydride phases suffer from a greatly reduced superconducting  $T_c$
  - Does the size of the precipitate matter? Large precipitates -> Q-disease; small precipitates, interstitial hydrogen -> Q-slope?
- Niobium hydride phase modeling showed
  - The driving forces for phase formation are evident in both lattice strain and energetic analyses of the presented models
  - Niobium lattice point defects can serve as nucleation centers for phase formation

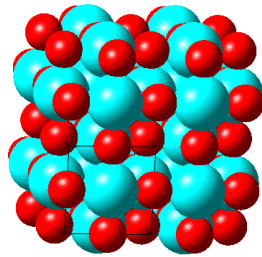
# Other Impurities in Niobium Oxide Layers

Nb – cubic



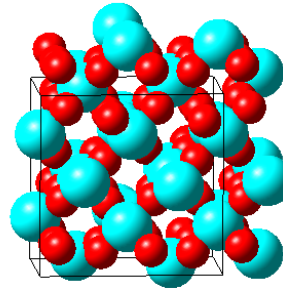
3.3 Å; 90°

NbO – cubic



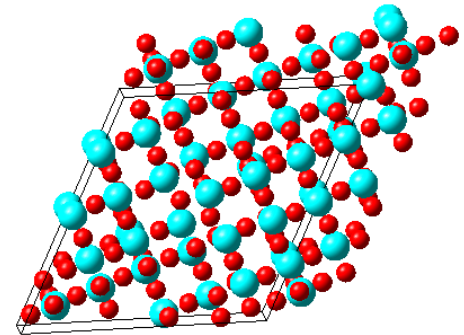
4.2 Å; 90°

NbO<sub>2</sub> – tetragonal



9.7, 9.7, 6.0 Å;  
90°

Nb<sub>2</sub>O<sub>5</sub> – monoclinic

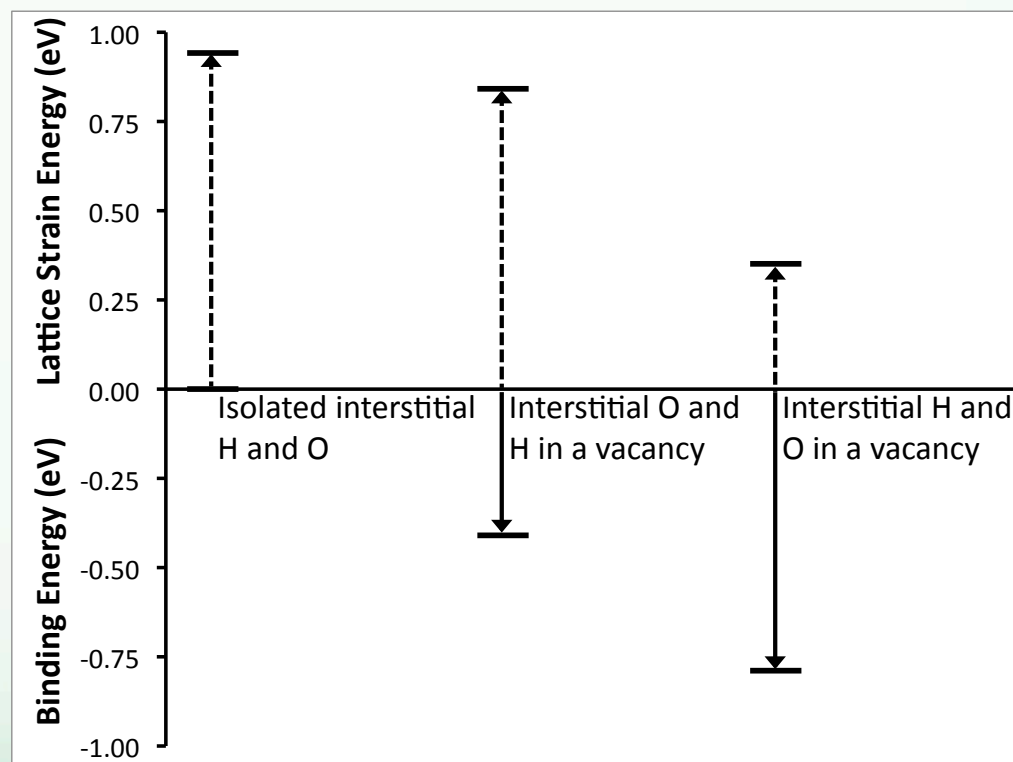
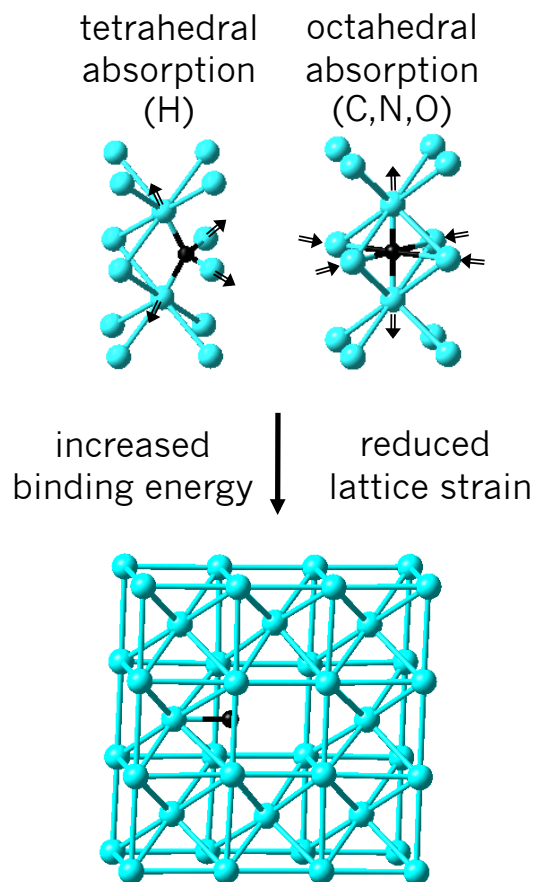


20.7, 3.8, 19.6 Å;  
90, 65, 90°

- Thermodynamic ordering of oxide layers on the niobium surface
- Strained and defective interfaces
- Surface layer can be highly defective or amorphous
- Interfaces can be trapping centers for other impurities such as H

# Other Impurities in Niobium

## Relation to Hydrogen



# Application to SRF Cavities

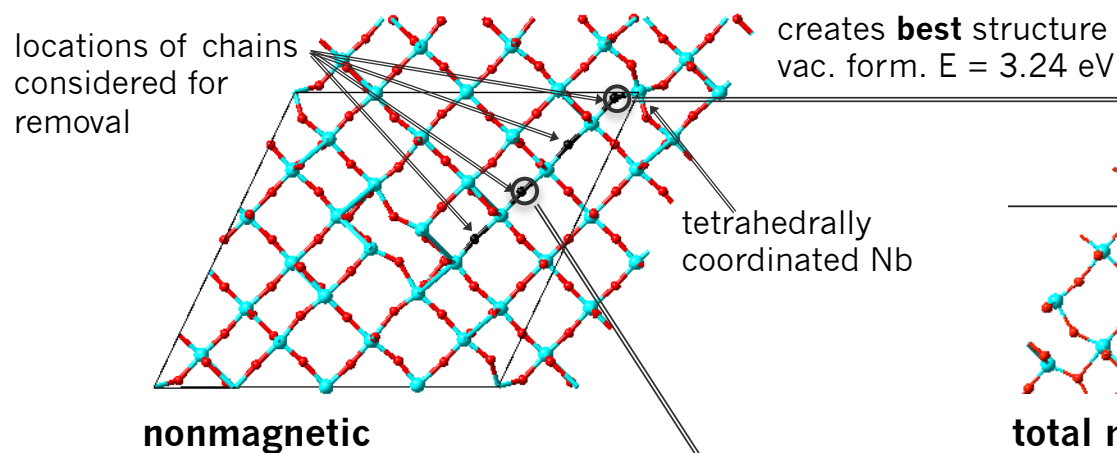
- The success of the 120 °C bake may be related to the interaction between impurities and lattice defects.
- Suggested mechanism:
  - > Hydrogen is liberated from both the ordered hydride phases and the niobium site vacancies
  - > Hydrogen diffuses into the niobium bulk
  - > Some oxygen diffuses from the oxide phases or niobium interstitial sites and becomes trapped by the niobium vacancies in the near surface region
  - > Hydrogen is prevented from returning to those phase nucleation centers
  - > Hydrogen may also become trapped by sites in the niobium bulk

# Oxygen in Niobium

## An Additional Concern

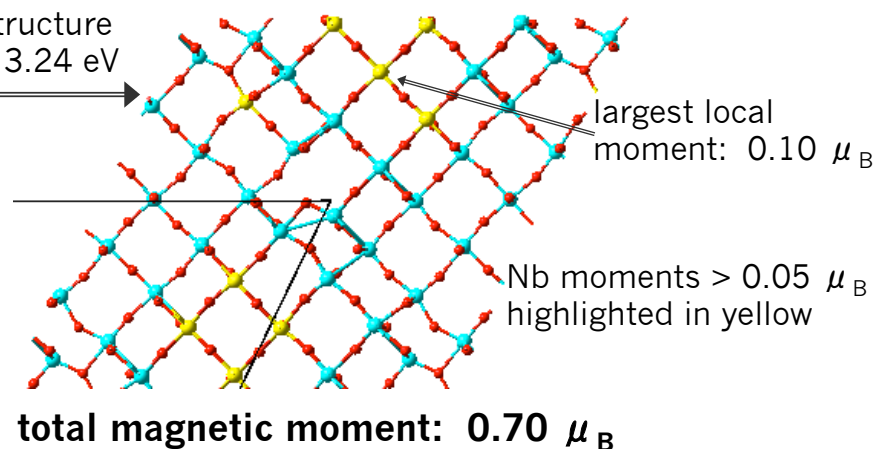
### Create a chain of oxygen vacancies (1.4%)

Beginning Structure – P2 Nb<sub>2</sub>O<sub>5</sub>



creates **worst** structure vac. form.  $E = 5.23$  eV  
**nonmagnetic**

Energetically Best Structure



# Summary

- Hydrogen in Niobium
  - Phase properties and driving forces for phase formation
  - Nucleation of phases by niobium lattice defects
    - > Removal of nucleation sites -> prevention of hydrides
- Other impurities (O, N, C) in Niobium
  - Oxide structures
  - Interactions between impurities and the niobium lattice during the 120 °C bake
    - > dissociation/prevention of hydrides
  - Magnetism in defective oxides

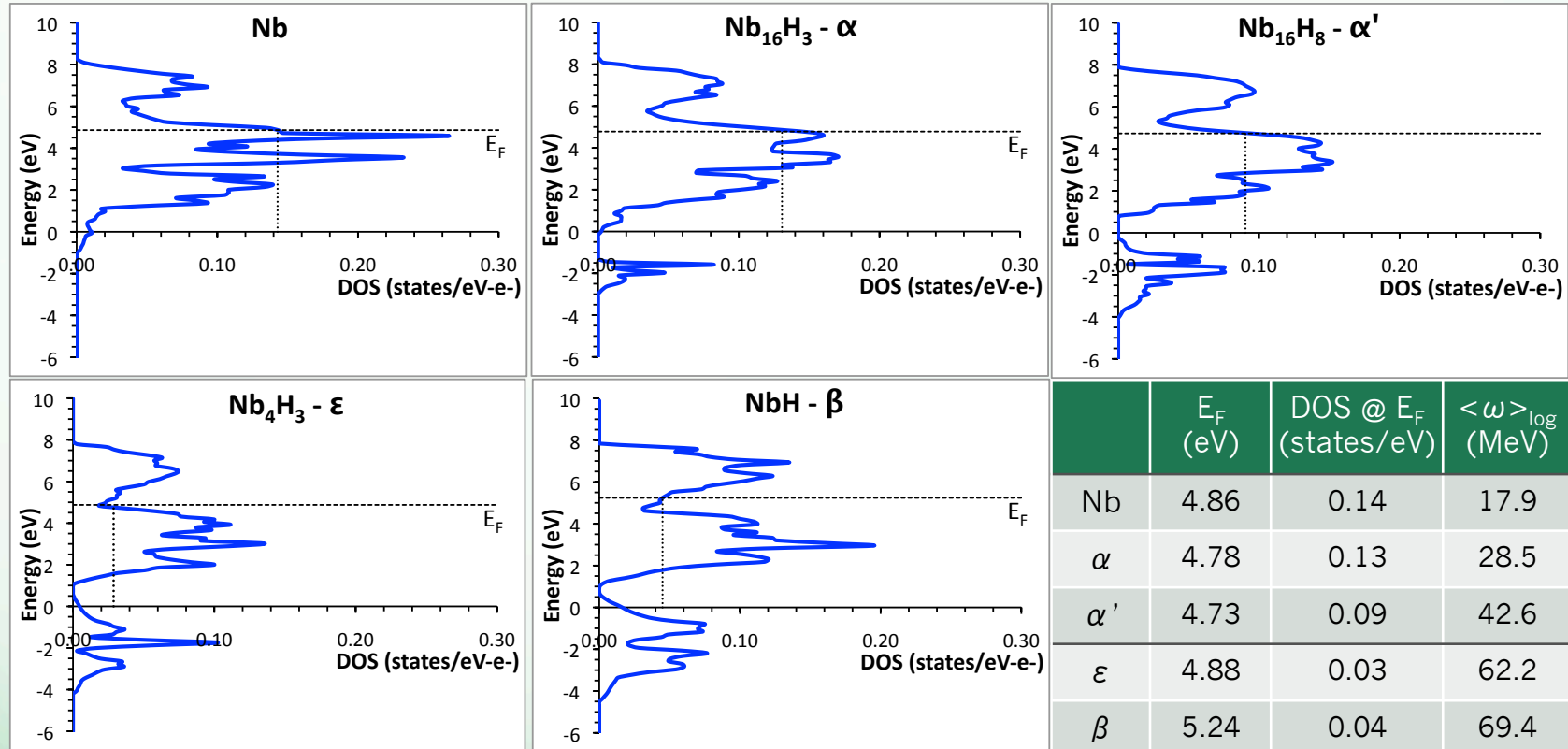
# Acknowledgments

- Lance Cooley and Superconducting Materials Group members
- David Seidman and group members
- Computing resources at Fermilab



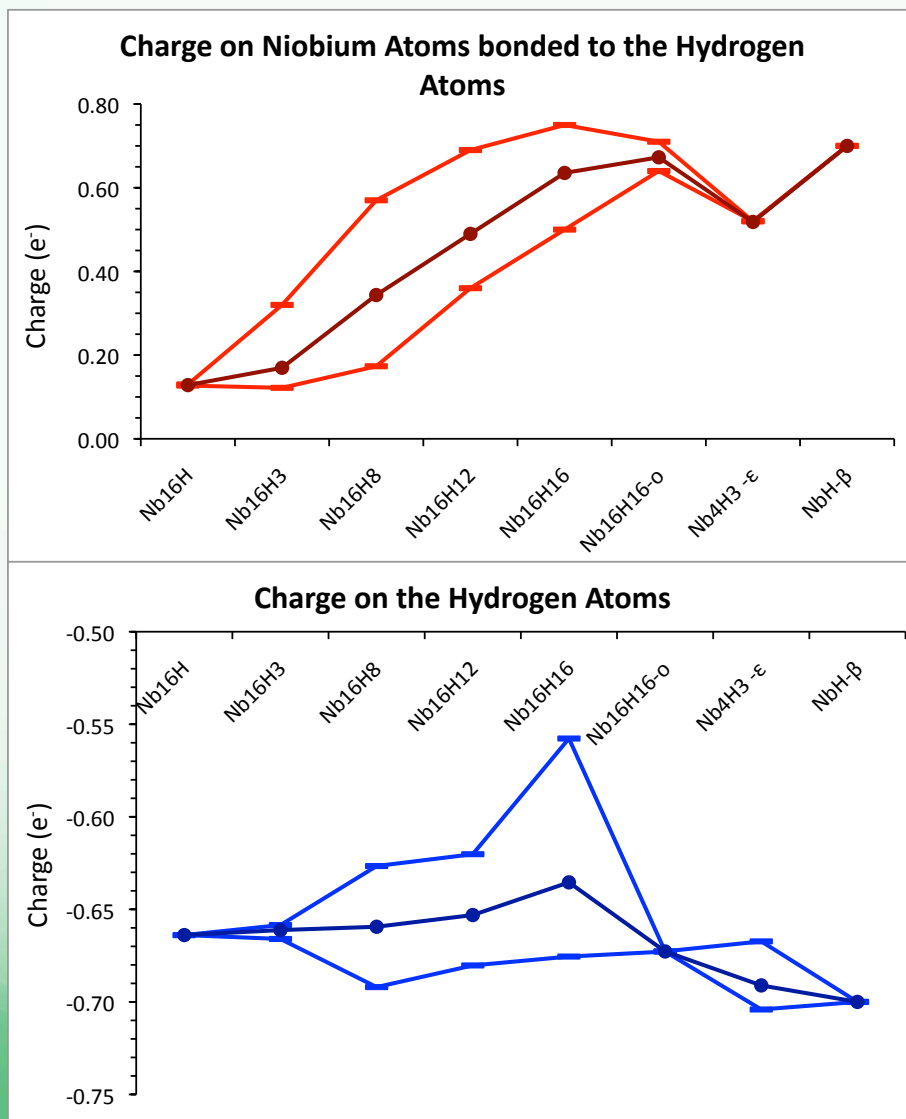
# Hydrogen in Niobium

## Electron and Phonon States



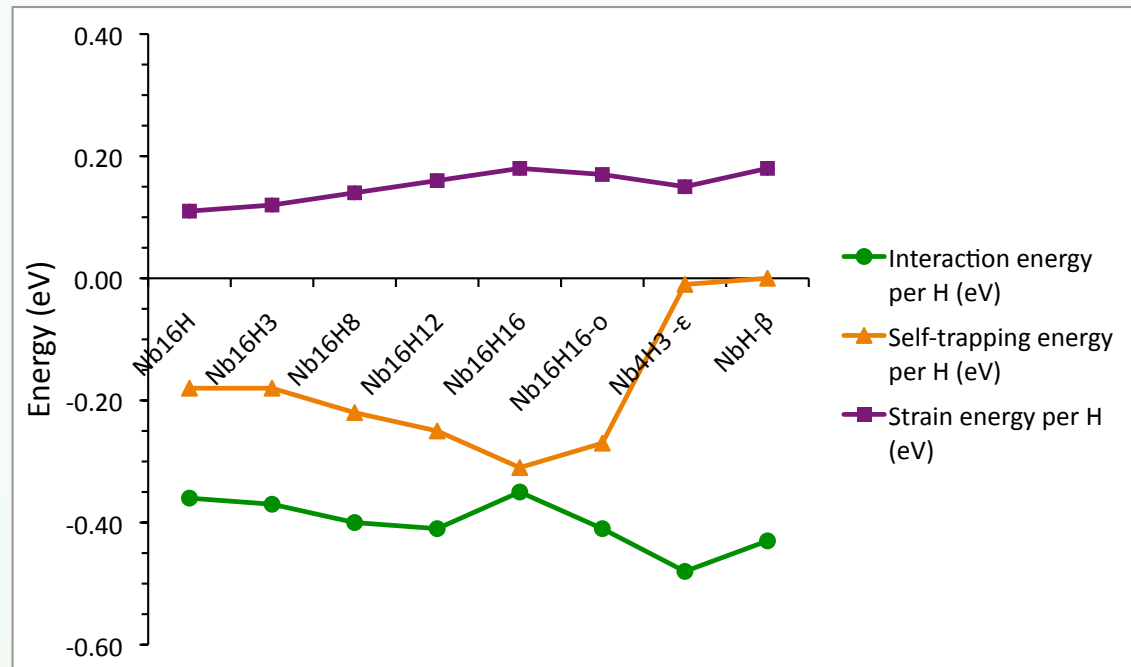
# Hydrogen in Niobium

## Local Charges



- The highest, lowest, and average charges are shown
- The charge on Nb atoms increases with increasing H concentration
- The charge on the H atoms slightly decreases with increasing H concentration and is the lowest for the ordered phases
- The charges on the Nb and H atoms are ~ constant for the ordered phases and equal and opposite for the beta phase

# Hydrogen in Niobium Energy Breakdown



- Interaction energy and self-trapping energy decrease with increasing H concentration; self-trapping energy is zero for the ordered configurations
- Lattice strain energy increases with increasing H concentration, is ~ equal for the disordered and ordered hydrogen configurations of the same concentration