

# **Low Field Quench, HAZ Pits and Strain in a Prototype Cavity for the European XFEL**

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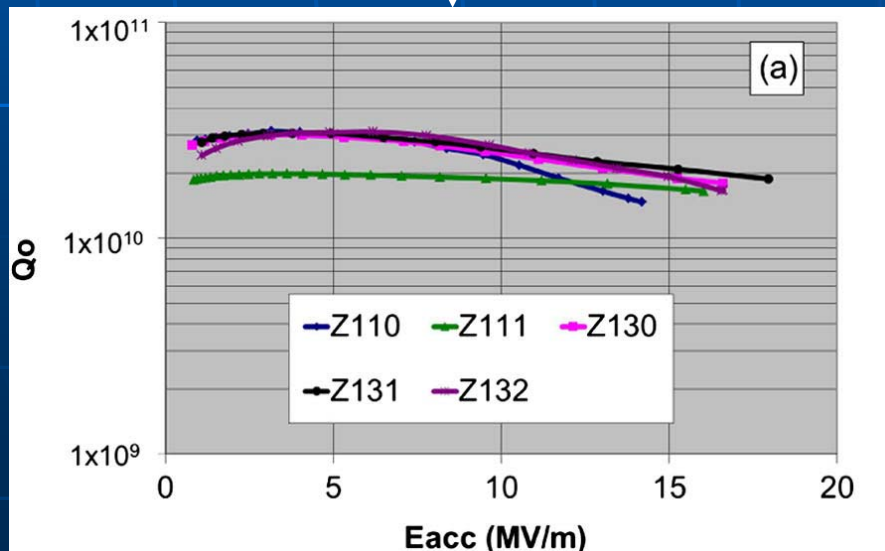
Analysis of results from:

W. Singer, X. Singer, S. Aderhold, A. Ermakov, and K. Twarowski, R. Crooks, M. Hoss, F. Schölz, and B. Spaniol; "Surface investigation on prototype cavities for the European X-ray Free Electron Laser," PHYSICAL REVIEW SPECIAL TOPICS - ACCELERATORS AND BEAMS 14, 050702 (2011).

Work funded by W. Singer, Deutsches Elektronen Synchrotron

# Overview

- Some SRF Cavities manufactured for the European XFEL showed thermal breakdown without field emission at low accelerating gradients
- Thermometry indicated hot spots in the equator weld region
- A hot spot region from one cavity, Z111 (**17 MV/m** quench), was removed by dry, slow speed, hole saw milling and examined by Scanning Electron Microscopy and Electron Backscattered Diffraction
- Comments on **Deep-drawing Strain, HAZ and Pits**
- Correlations were made between *stored strain energy and pits in hot spot sample*, use of GROD for relative dislocation density



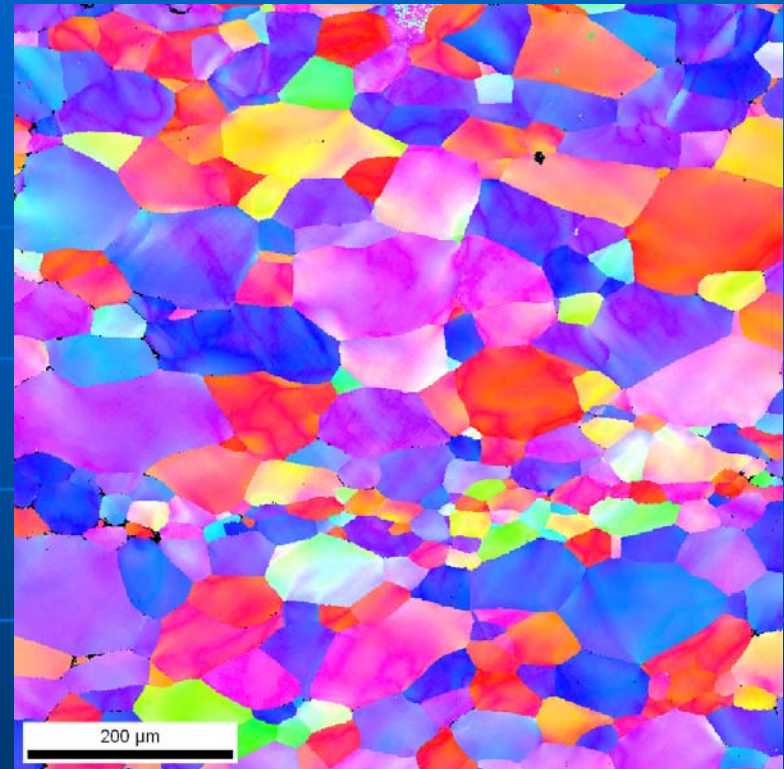
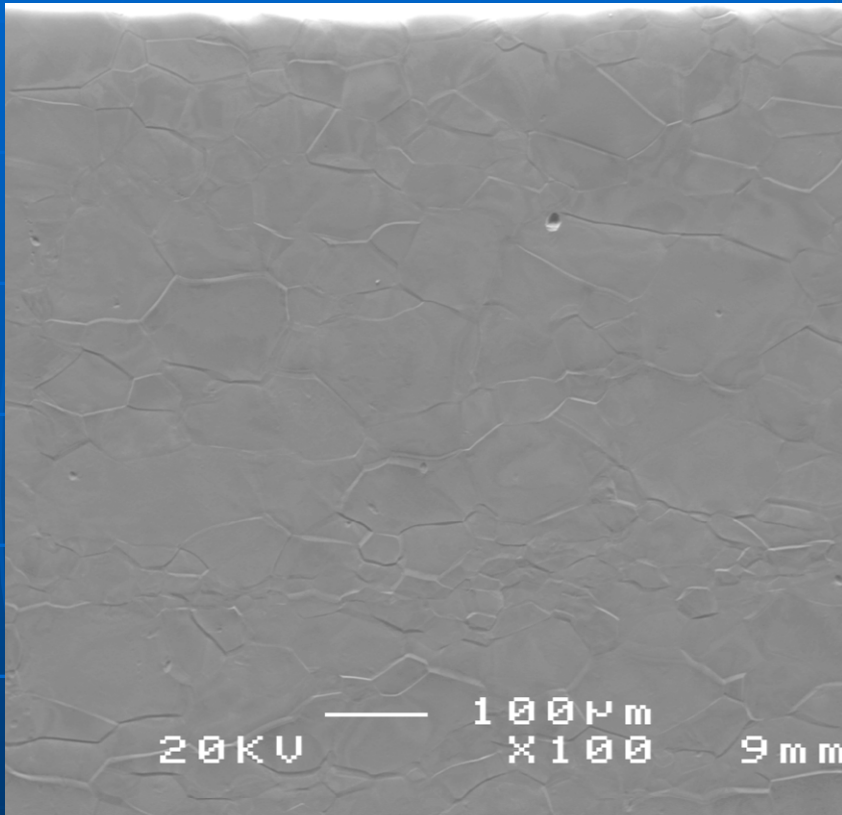
## Treatment of Z111:

- > 100  $\mu\text{m}$  removed by EP
- $\sim 10 \mu\text{m}$  removed by BCP
- 120°C bake
- rf testing at DESY with thermometry
- hot spot removal (X. Singer)



# Background: Deep Drawing Strain

inside surface



~ 20% strain

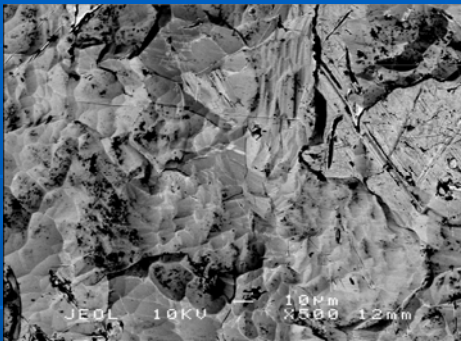
Mark III CEBAF  
Cavity

95% hit rate, 560,200 data points,  
1 µm spatial resolution



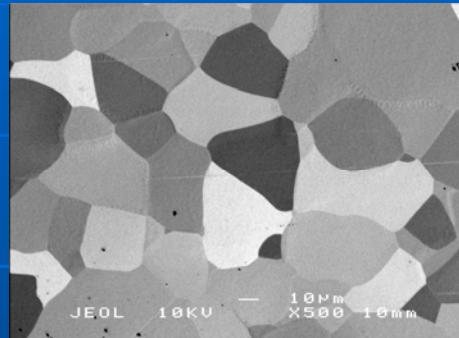
# Background: AR EP Sheet Compared to Deep Drawing Strain at Equator

AR



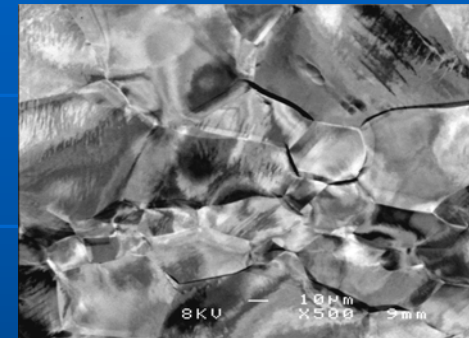
BEI

EP



BEI

Deep-drawn Cavity Equator



BEI

High strain removed by welding?






How much strain?  
lattice curvature  $\sim 1^\circ/\mu\text{m}$




$\sim 20\%$  strain

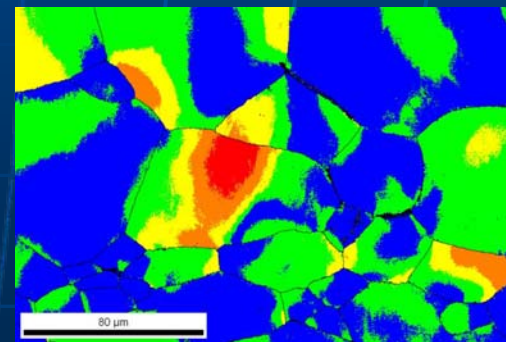
EBSD GROD Map 500  $\mu\text{m}$  from interior,  
3 mm from lip  
0.3  $\mu\text{m}$  spatial resolution

Color Coded Map Type: Grain Reference Orientation Deviation

	Min	Max	Total Fraction	Partition Fraction
	0	3.87436	0.488	0.488
	3.87436	7.74872	0.369	0.369
	7.74872	11.6231	0.088	0.088
	11.6231	15.4974	0.038	0.038
	15.4974	19.3718	0.013	0.013

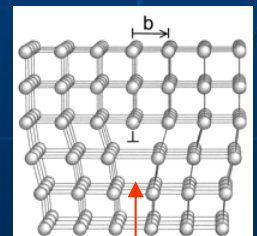
Boundaries: Rotation Angle

	Min	Max	Fraction	Number	Length
	15°	180°	0.011	8093	2.43 mm



EBSD

Edge Dislocation



interstitial site



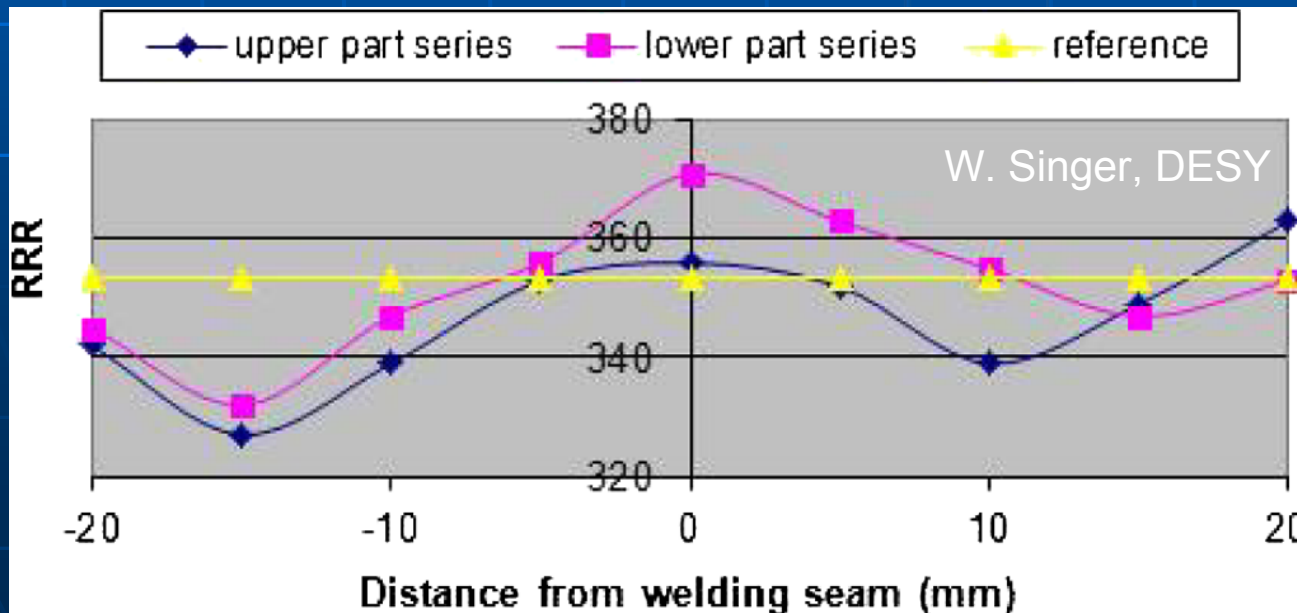
# Background: Equator welds, HAZ pits and RRR drop



HAZ exposed to high T

Quench related to topography (pits)?

K. Watanabe, *Recent Inspection Results by Kyoto-Camera*, in *TESLA Technology Collaboration Meeting*. 2008: New Delhi, India.



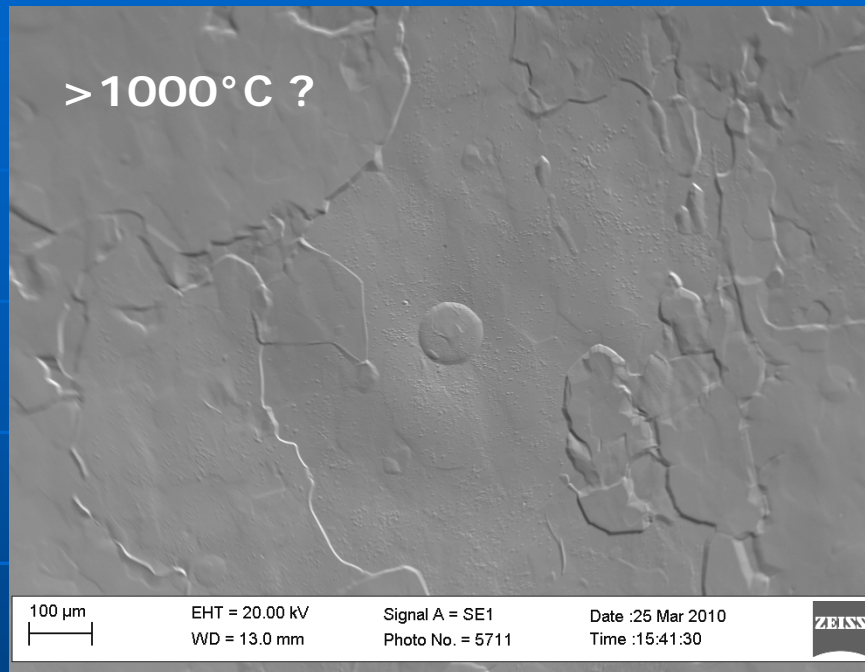
Low RRR related to HAZ?

Diffusion of interstitials to dislocation sinks?  
During welding?  
During ep?

Oxygen lowers  $T_c$  and  $H_c$ .

Other impurities?

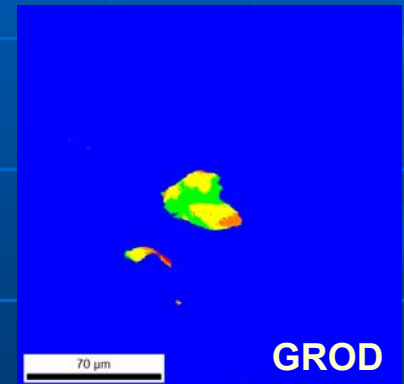
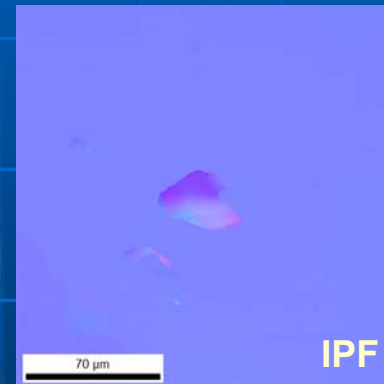
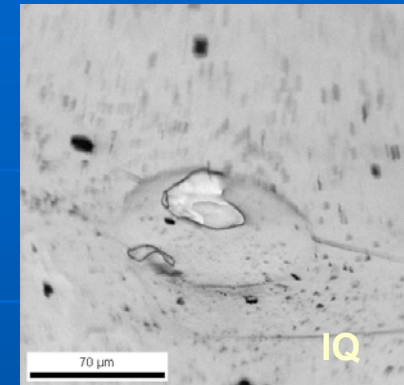
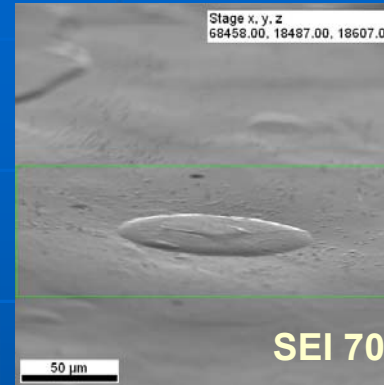
# HAZ pit and “pocket of strain” in Z111



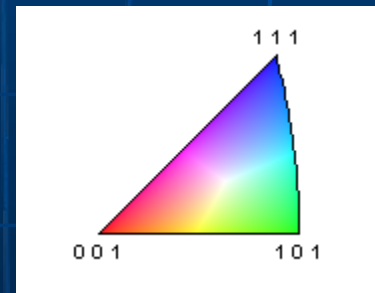
Pit about 2 mm within HAZ

Strain region part of larger grain,  
no high angle g.b.  $> 15^\circ$

**How much strain?**  
**lattice curvature  $\sim 3^\circ/\mu$ m**  
**(greater than as- formed data)**



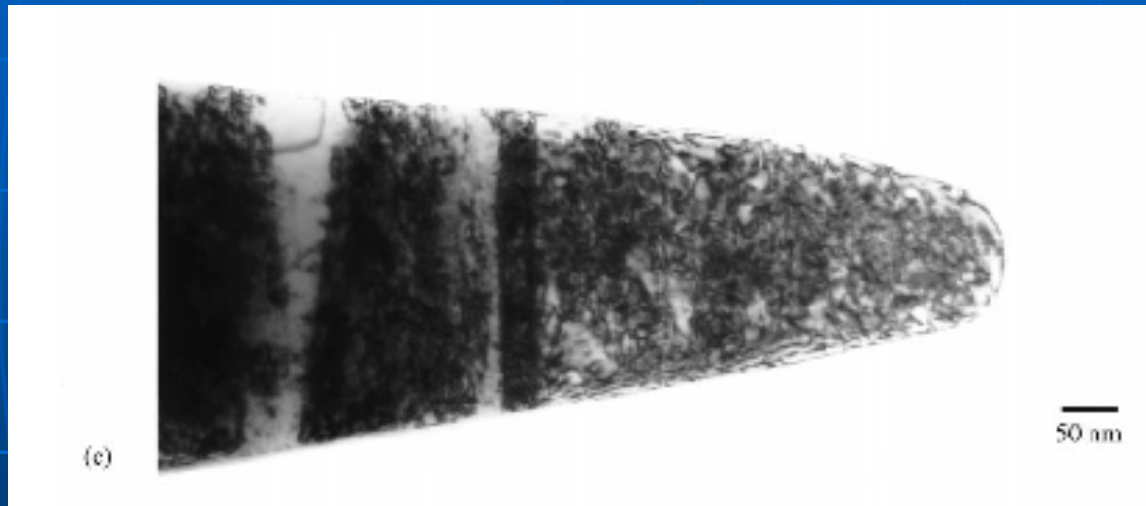
200 x 200  $\mu$ m, 0.5  $\mu$ m resolution



	Min	Max	Total Fraction
Blue	0	2.0103	0.977
Green	2.0103	4.02061	0.009
Yellow	4.02061	6.03091	0.011
Orange	6.03091	8.04122	0.002
Red	8.04122	10.0515	0.000

# 3DAP Sample (Fe alloy)

high dislocation density maintained

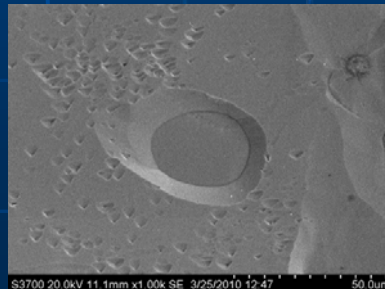


J. Wilde, et al., Scripta materialia, 43 (2000) 39.

# Conclusions and Conjecture

- Correlation found between low Eacc quench, hot spot, pit and strain
- Pocket of high dislocation density within pit in HAZ
  - “left behind” in ( $> 1000^{\circ}\text{C}$ ) recrystallization of HAZ
  - dislocations are sinks for interstitials
    - high resolution (3DAP?) study appropriate to verify dislocation atmospheres, interstitial concentration, chemistry
- Do dislocation atmospheres stabilize dislocations and inhibit recrystallization?
  - questions of diffusion, recovery and recrystallization rates during a brief thermal excursion
- Hot spot due to impurity concentration rather than topography?

> 100  $\mu\text{m}$  removed by EP  
~10  $\mu\text{m}$  removed by BCP  
120°C bake



~ 80  $\mu\text{m}$  diameter  
~ 10  $\mu\text{m}$  deep